

**Comparative study on microstructures and properties of  
 $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4\text{-C}$  ceramic filter and  $\text{Al}_2\text{O}_3\text{-C}$  ceramic filter**

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— PART ONE —



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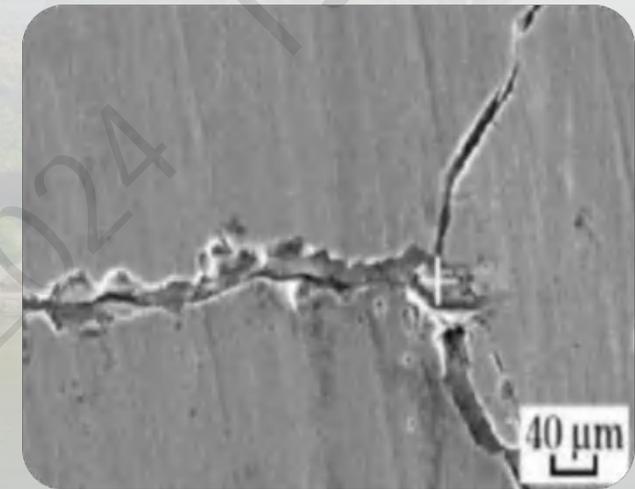
# Research background



# Research background

Under the background of “Made in China 2025” policy, the demand for **ultra-clean steel in major equipment manufacturing and major engineering construction** in China is increasing day by day.

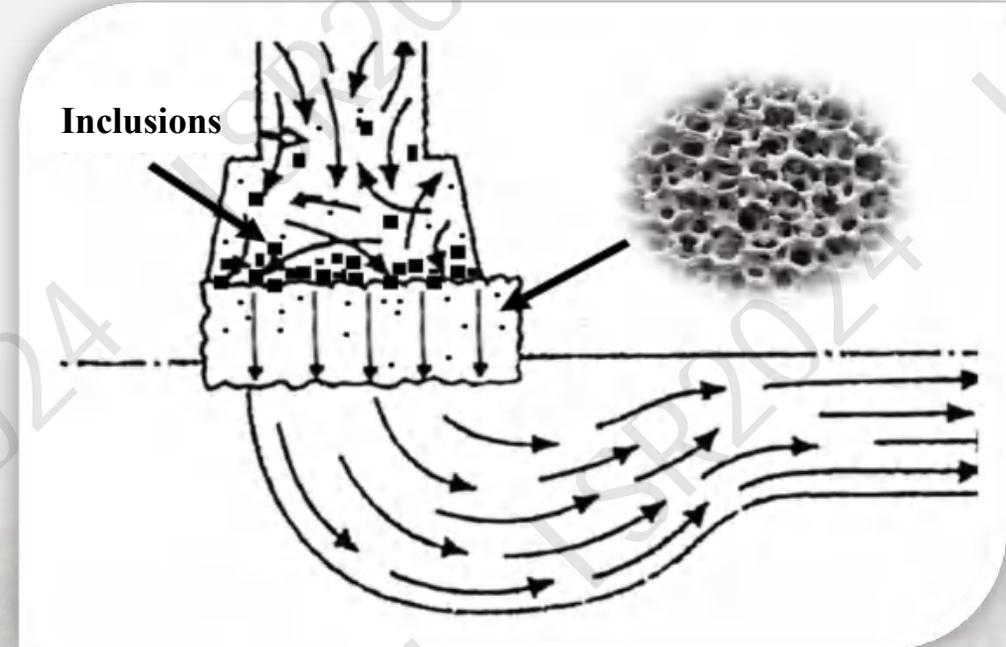
**Non-metallic inclusion** in steel is one of the important factors that **affects steel properties and safety applications**. Therefore, **how to reduce the non-metallic inclusions and improve the quality of steel** is a problem that needs to be solved at present.





# Research background

**The porous ceramic filter** has a three-dimensional network structure, a porosity of up to 70-90%, and a large surface area, mainly used in the final procedure of steel casting, **which is one of the effective ways to adsorb and filter the non-metallic inclusions in molten steel.**



**CaO filter**

→ Easy hydration

**SiC filter**

→ Low filtration efficiency

**Zr<sub>2</sub>O filter**

→ Poor volume stability

**Al<sub>2</sub>O<sub>3</sub> filter**

→ Poor thermal shock resistance

**Al<sub>2</sub>O<sub>3</sub>-C filter**

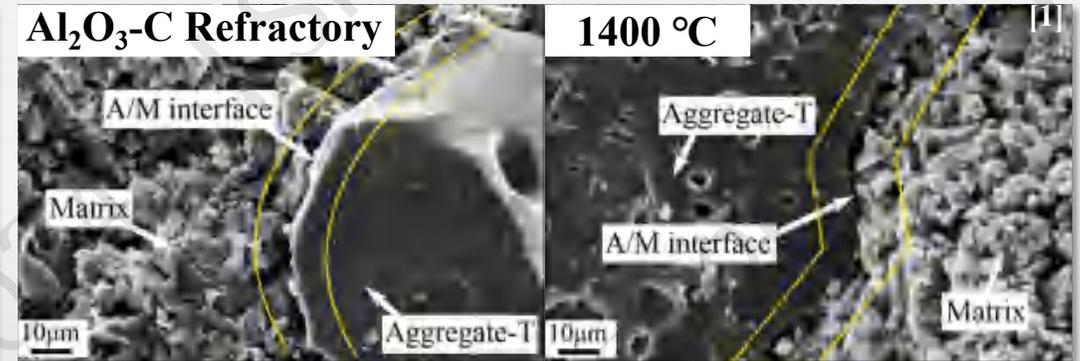
→  **Excellent filtration performance, thermal shock resistance and creep resistance**



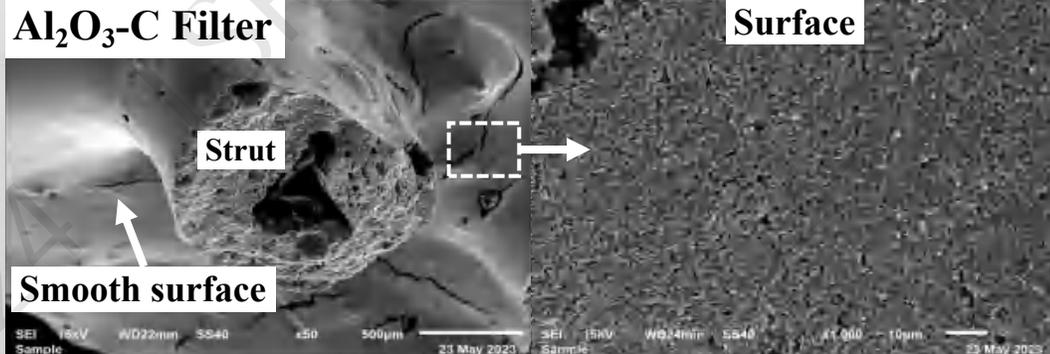


## The inherent limitations of the $\text{Al}_2\text{O}_3\text{-C}$ materials

① In the aspect of sintering of strut, **carbon is an inert phase in sintering theory**, and **it is difficult to form a strong neck bond between carbon and  $\text{Al}_2\text{O}_3$** , limiting the further enhancement of thermal shock resistance.



**Limit the further improvement of thermal shock resistance and strength!**



② In the other aspect of adsorption of inclusions, the **smooth surface of dense  $\text{Al}_2\text{O}_3$  leads to a small contact area and contact angle with molten steel**, thereby limiting the further improvement of inclusion adsorption and purification efficiency.

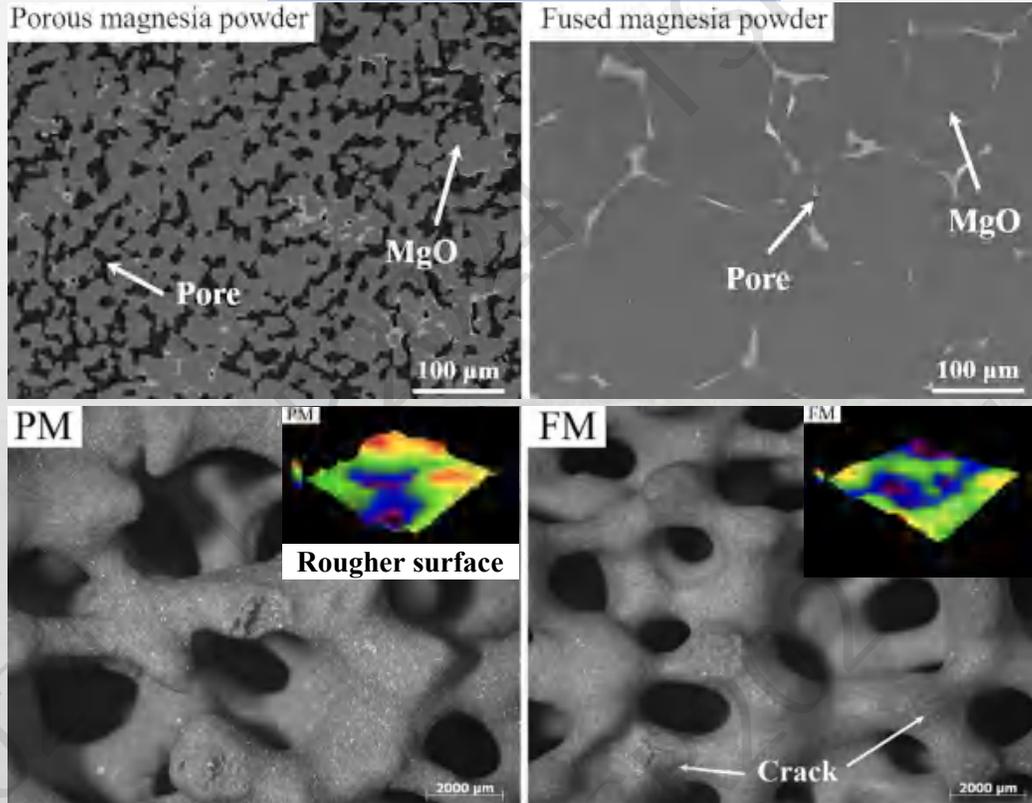
**Limit the further improvement of inclusion adsorption and purification efficiency!**



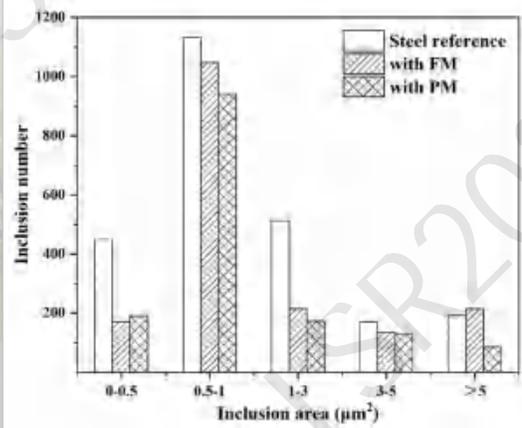
# Research background

**Firstly, pore is an important component of the microstructure.**

## MgO ceramic filter



Physical properties	PM	FM
Bulk density of filter (g/cm <sup>3</sup> )	0.91 ± 0.01	0.88 ± 0.03
Apparent porosity of filter (%)	74.0 ± 0.01	74.1 ± 0.03
CCS (MPa)	<b>1.71 ± 0.11</b> ↑UP	1.46 ± 0.09
CCS <sub>TS</sub> (MPa)	<b>1.32 ± 0.08</b>	1.08 ± 0.05
Retention rate of CCS (%)	<b>77.2</b>	74.0



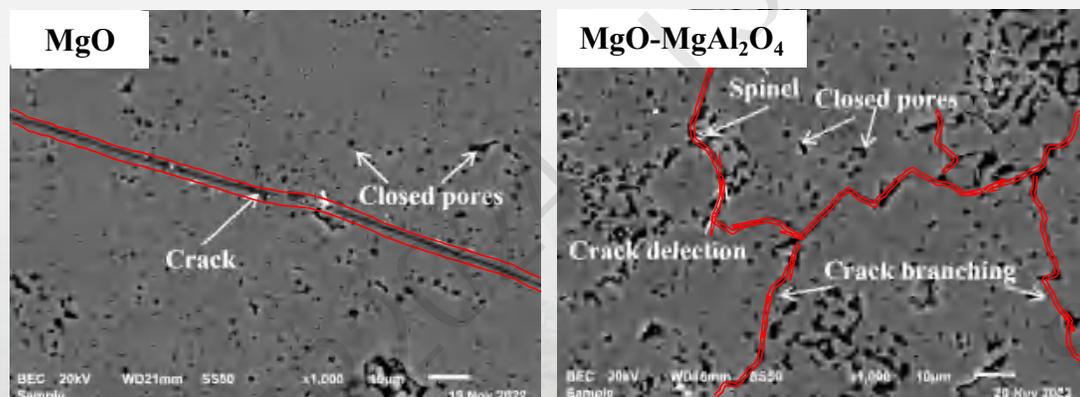
Chemical compositions of the steel samples	C (wt%)	Al (wt%)	T.O (ppm)
Steel reference	0.100	0.250	37.5
with FM	0.054	0.084	33.1
<b>with PM</b>	<b>0.054</b>	<b>0.072</b>	<b>29.4</b>



# Research background

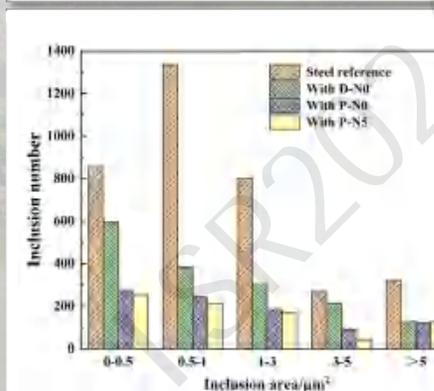
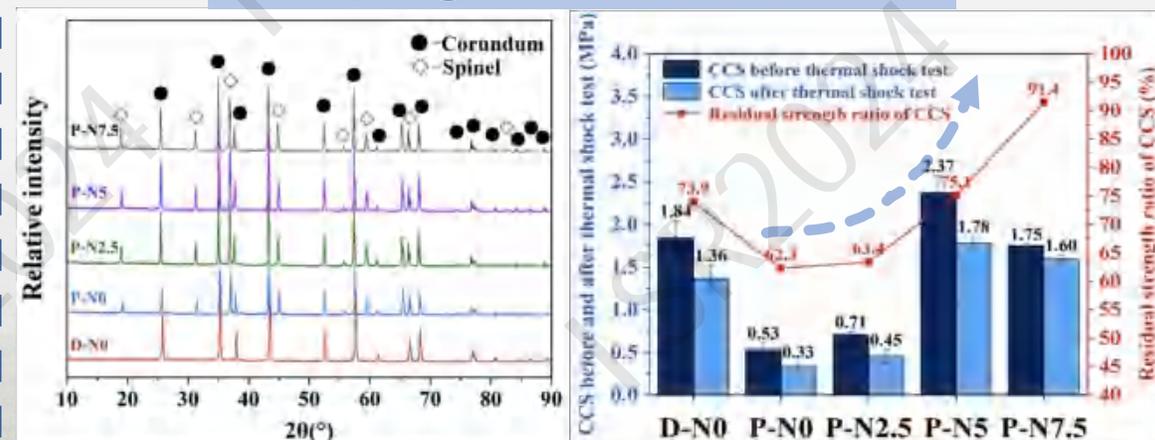
Secondly,  $MgAl_2O_4$  is a crucial phase for enhancing the thermal shock resistance of  $MgO-Al_2O_3$  system materials.

## MgO- $MgAl_2O_4$ ceramic filter



Physical properties	MA0	MA5	MA10	MA15	MA20
Bulk density of strut ( $g/cm^3$ )	2.52	2.77	2.85	2.87	2.96
Apparent porosity of strut (%)	28.8	22.3	20.4	20.1	17.0
CCS (MPa)	<b>1.68</b>	<b>2.04</b>	1.78	1.84	<b>2.35</b>
CCS <sub>TS</sub> (MPa)	<b>1.29</b>	<b>1.93</b>	1.56	1.62	<b>1.91</b>
Retention rate of CCS (%)	<b>76.7</b>	<b>94.6</b>	87.6	88.0	<b>81.3</b>

## $Al_2O_3$ - $MgAl_2O_4$ ceramic filter



## Chemical compositions of the steel samples

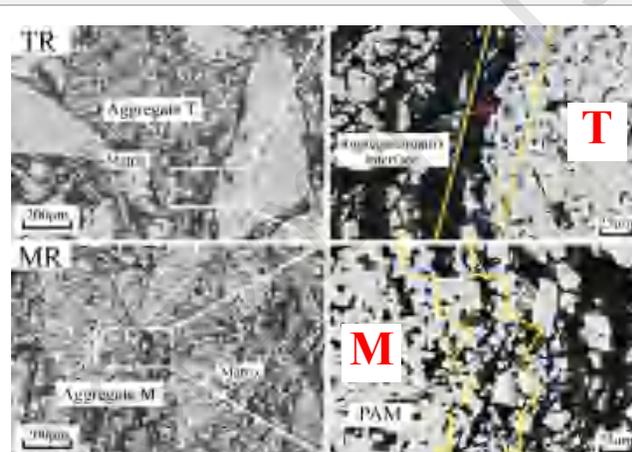
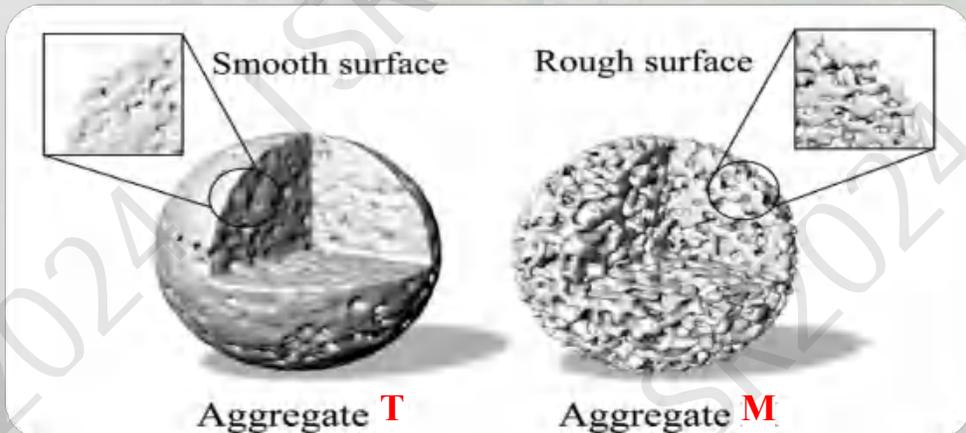
	Al (wt%)	T.O (ppm)
Steel reference	0.46	40.2
with D-N0	0.25	21.0
with P-N0	0.18	16.6
<b>with P-N5</b>	<b>0.18</b>	<b>12.7</b>



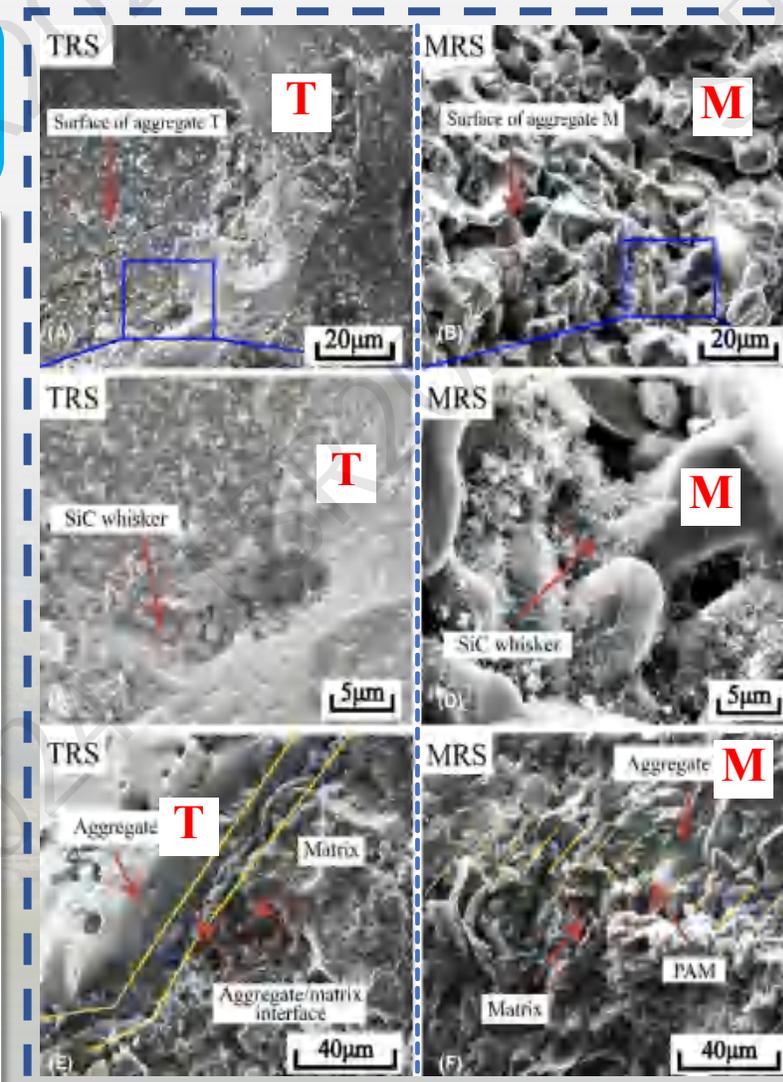
# Research background

Thirdly, the **synergistic effect** of the **porous structure** and  **$\beta$ -SiC whiskers** is an effective approach to enhancing the **mechanical properties** of  $\text{Al}_2\text{O}_3$ -C materials.

Physical properties	Aggregate T	Aggregate M
Bulk density ( $\text{g}/\text{cm}^3$ )	$3.54 \pm 0.02$	$2.82 \pm 0.02$
True density ( $\text{g}/\text{cm}^3$ )	$3.904 \pm 0.003$	$3.967 \pm 0.003$
Apparent porosity (%)	$6.2 \pm 1$	$25.7 \pm 1$
Closed porosity (%)	$3.1 \pm 1$	$3.2 \pm 1$
Total porosity (%)	<b><math>9.3 \pm 1</math></b>	<b><math>28.9 \pm 1</math></b>

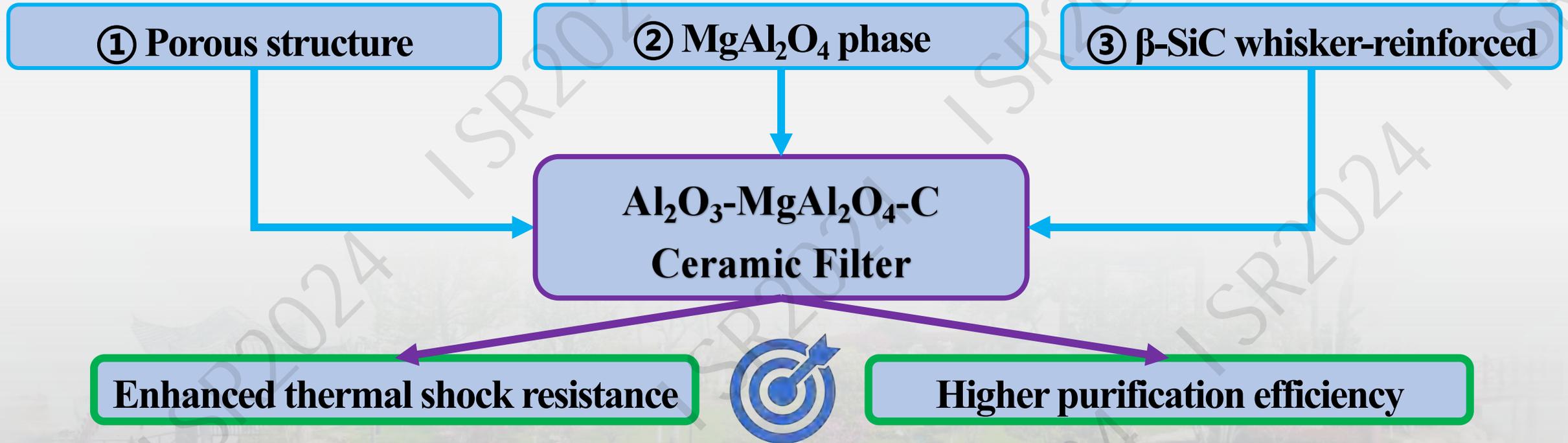


Physical properties of refractories	TRS	MRS
Bulk density ( $\text{g}/\text{cm}^3$ )	2.87	2.66
Apparent porosity (%)	17.4	23.3
<b>CMOR (MPa)</b>	<b>18.2</b>	<b>23.7</b>





# Research background



We propose the utilization of **porous  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  micropowder**, high-reactive **nano-carbon black**, and **Si powder** to prepare the  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4\text{-C}$  filter. This approach aims to **enhance thermal shock resistance through the synergistic toughening effects of  $\beta\text{-SiC}$  whiskers and the  $\text{MgAl}_2\text{O}_4$  phase**, while also **improving the purification efficiency for molten steel by increasing the surface roughness of the filter's porous struts**.

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— PART TWO —



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# Experiment procedures



# Experiment procedures

## Composition of the slurries (wt%)

		AM	A
Solids	<b>Porous Al<sub>2</sub>O<sub>3</sub>-MgAl<sub>2</sub>O<sub>4</sub> micropowder</b>	<b>80</b>	0
	$\alpha$ -Al <sub>2</sub> O <sub>3</sub> micropowder AMA-10	0	80
	Si	6	6
	Carbores®P	6	6
	Nano-Carbon Black N220	8	8
Additives <sup>a</sup>	Carboxymethyl cellulose C804619	0.36	0.36
	Lignosulfonate C804808	1.5	1.5
	Polycarboxylate WSM-M	0.3	0.3
	Glydol N1055	0.3	0.3
	Contraspum K1012	0.1	0.1

<sup>a</sup> Related to 100wt% solids

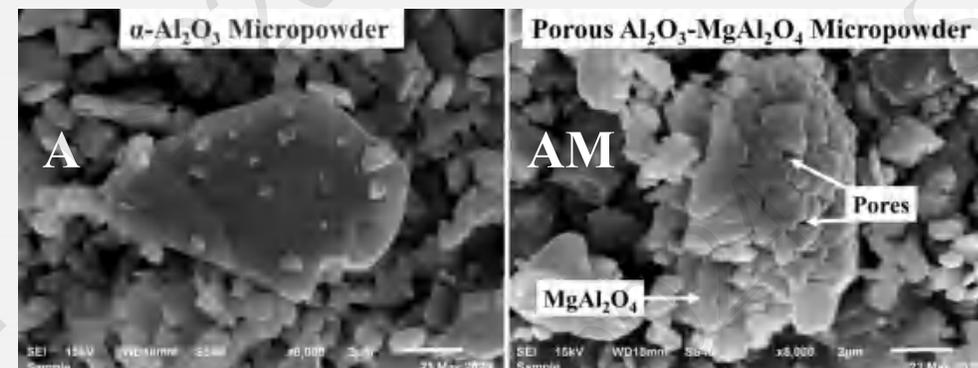


## Porous Al<sub>2</sub>O<sub>3</sub>-MgAl<sub>2</sub>O<sub>4</sub> raw material

Bulk density: 2.53 g/cm<sup>3</sup>; Apparent Porosity: 33.6%

Prepared by **in situ decomposition pore-forming technique**<sup>[8-10]</sup>.

The filters prepared with **porous Al<sub>2</sub>O<sub>3</sub>-MgAl<sub>2</sub>O<sub>4</sub> micropowder** were named **AM**.



## Chemical composition of the raw materials (wt%)

	$\alpha$ -Al <sub>2</sub> O <sub>3</sub> micropowder	Porous Al <sub>2</sub> O <sub>3</sub> -MgAl <sub>2</sub> O <sub>4</sub> micropowder
Al <sub>2</sub> O <sub>3</sub>	99.31	90.01
MgO	0.01	<b>8.60</b>
SiO <sub>2</sub>	0.25	0.33
CaO	0.08	0.15
Fe <sub>2</sub> O <sub>3</sub>	0.07	<b>0.41</b>
K <sub>2</sub> O	0.01	0.02
Na <sub>2</sub> O	0.05	0.15
TiO <sub>2</sub>	0.01	0.04
IL	-	0.12

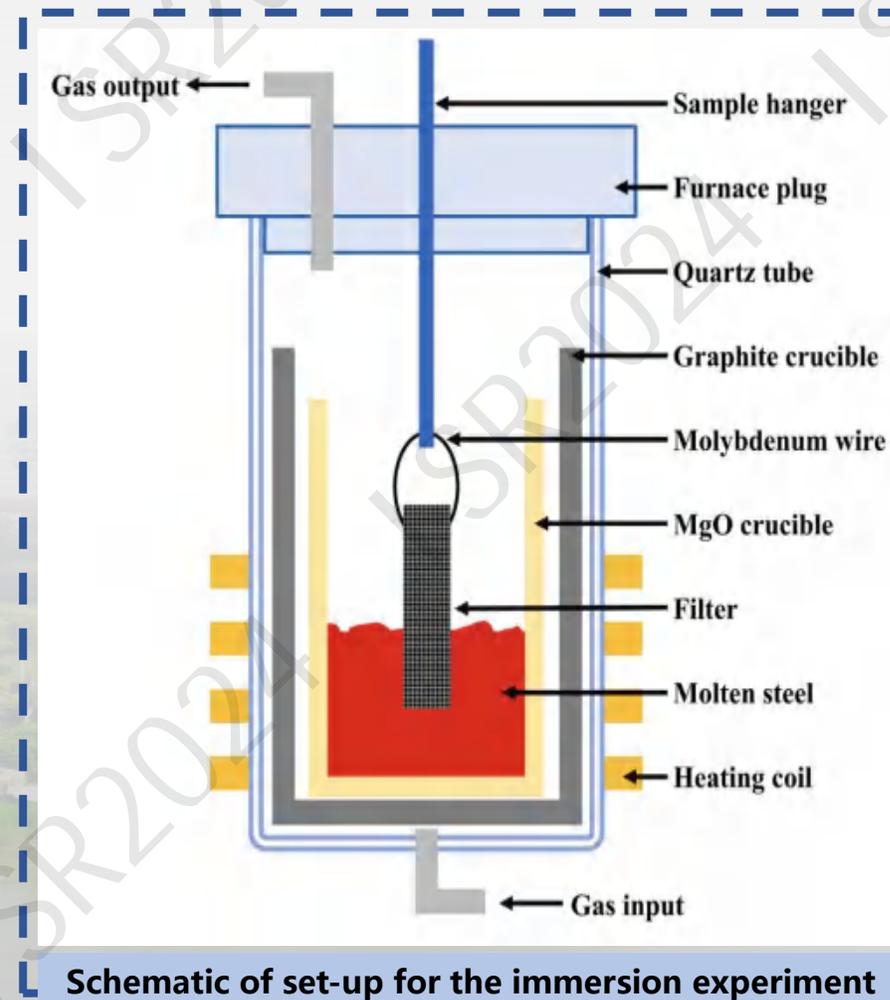


# Experiment procedures

## Immersion experiment

Al-killed steel (Wuhan Iron & Steel Co., China) was used in the immersion tests with a melting point of approximately 1540 °C and a composition of **C 0.063 wt%**, **Al 0.46 wt%**, **Si 0.017 wt%**, **S 0.017 wt%**, **Mn 0.051 wt%**. 200g of steel was used for each immersion experiment.

**The immersion experiment was performed in a high-frequency induction furnace in a rotating magnetic field.** The high purity **argon** with a purity of 99.99 % was slowly introduced with a flow rate of 0.5 L/min. When the temperature reached **1600 °C**, the filter was slowly immersed in the molten steel for **20 min** before the filter was separated from the molten steel and furnace-cooled.



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— PART THREE —

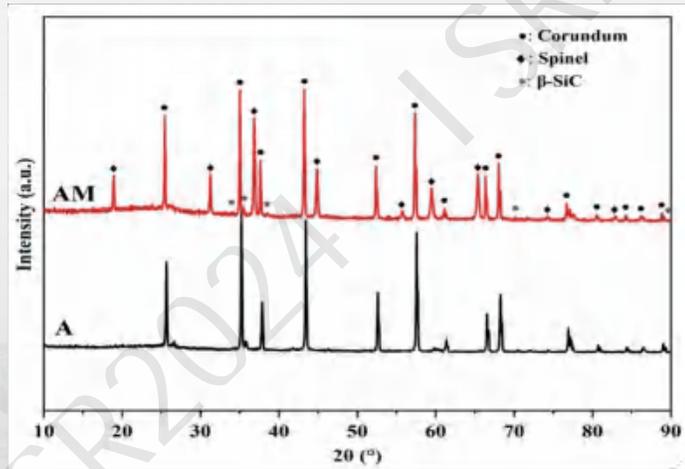


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# Results and discussion



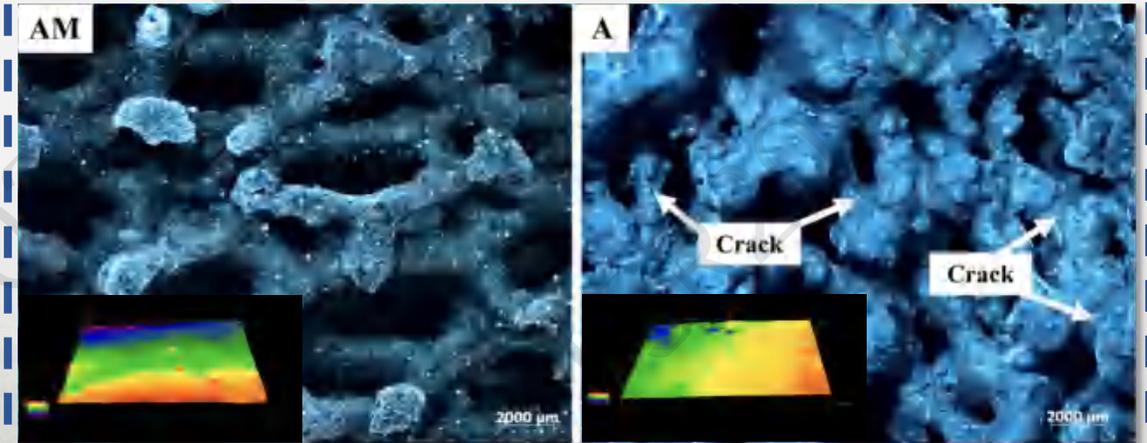
## Phase compositions (XRD results)



**Relative contents of phase in the specimens (wt%)**

	Corundum	Spinel	β-SiC
AM	58	33	9
A	97	-	3

## Morphologies of the filters

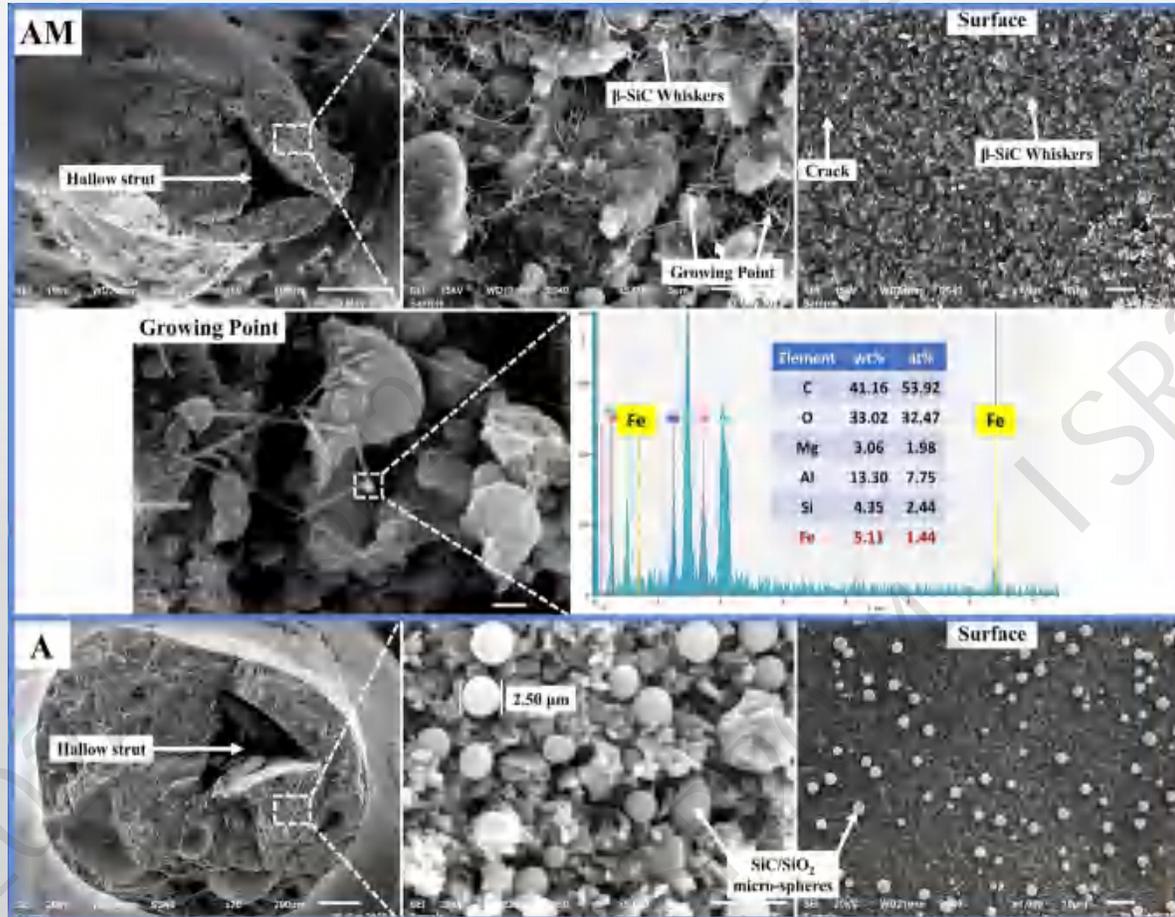


**Surface roughness parameters of the substrates AM and A**

	$R_q/\mu\text{m}$	$R_a/\mu\text{m}$	$R_z/\mu\text{m}$	$R_{SA}$
AM	0.799	0.626	3.055	1.072
A	0.393	0.303	1.520	1.021



## Microstructures of the filters



## Physical properties of the filters

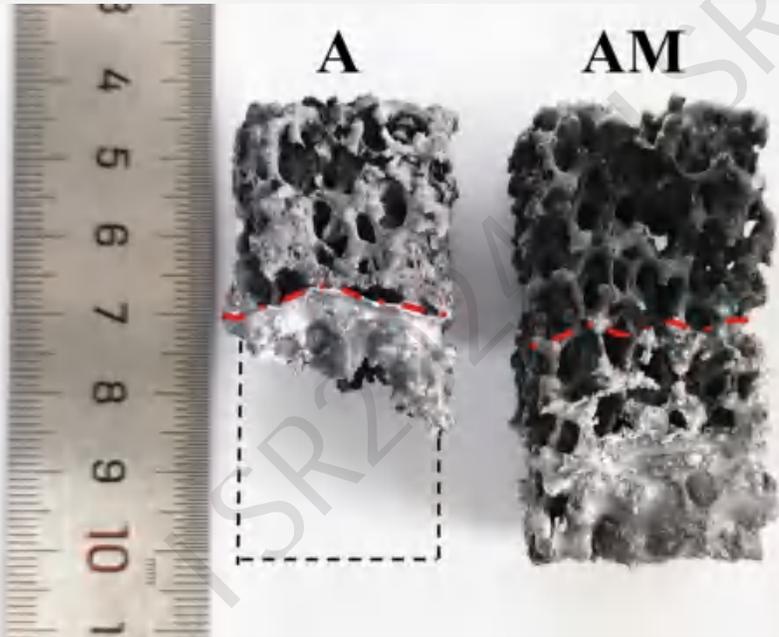
	AM	A
Bulk density of filter (g/cm <sup>3</sup> )	0.76 ± 0.05	0.89 ± 0.08
Porosity of filter (%)	57.30 ± 2.81	43.31 ± 5.09
CCS (MPa)	0.45 ± 0.10	0.62 ± 0.03
CCS <sub>TS</sub> (MPa)	0.41 ± 0.12	0.36 ± 0.13
Retention rate of CCS (%)	91.11	58.06

Filter AM exhibited a **lower bulk density** and a **higher porosity**. The cold compressive strength of filter A was 37.78% higher than that of filter AM. However, **after three thermal shock tests, filter AM demonstrated a higher cold compressive strength** than that of filter A. Specifically, **filter AM retained 91.11%** of its original strength, while filter A only retained 58.06%.

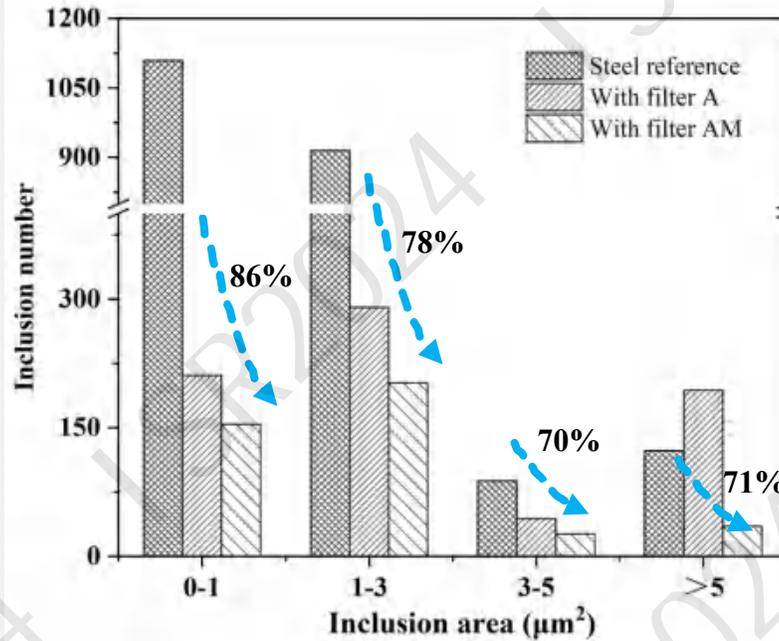


# Results and discussion

## Interaction between the filters and molten steel



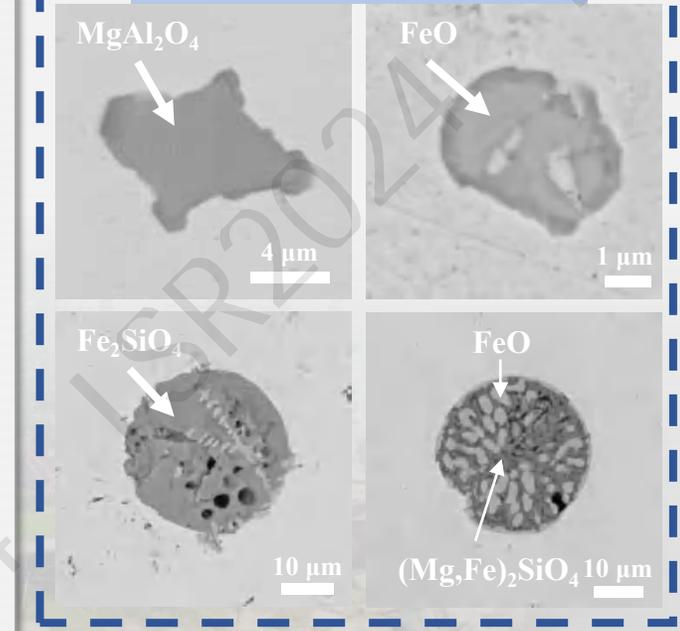
After immersion in the molten steel, **filter AM retained integrity**. Conversely, **filter A experienced a complete fracture**, with one half of the fractured specimen falling into the molten steel.



Chemical compositions of the steel specimens

	C (wt%)	Al (wt%)	Si (wt%)	T.O (ppm)
Steel reference	0.063	0.46	0.017	56.3
<b>with AM</b>	<b>0.071</b>	<b>0.24</b>	<b>0.078</b>	<b>11.7</b>
with A	0.47	0.29	0.16	52.2

## Inclusions



Filter AM reduced the T.O content from **56.3 ppm to 11.7 ppm**.

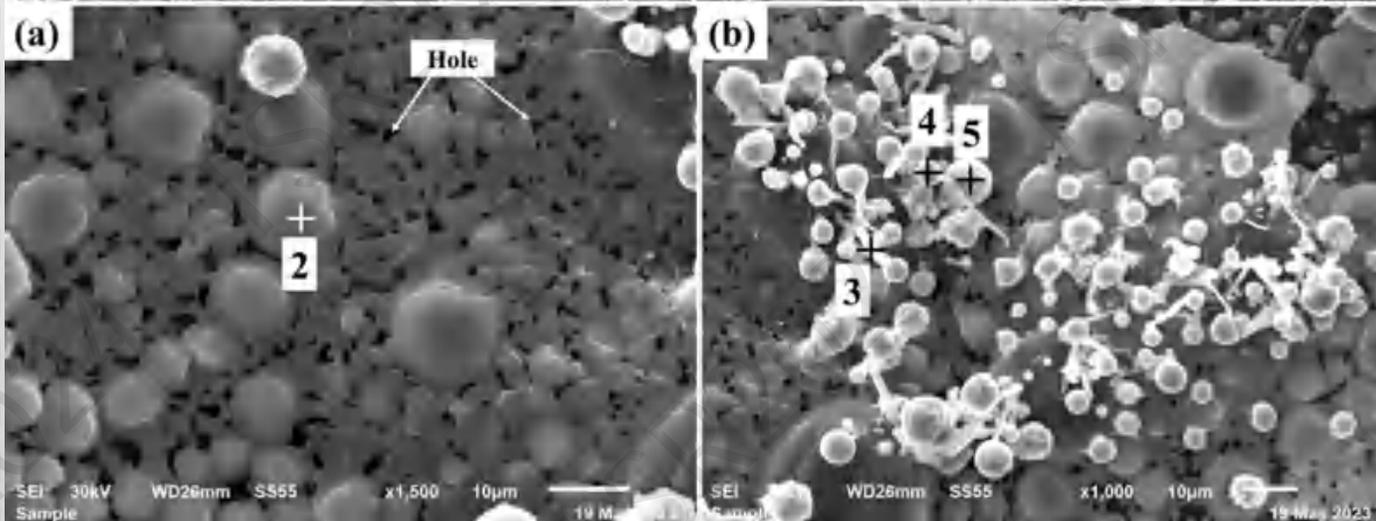
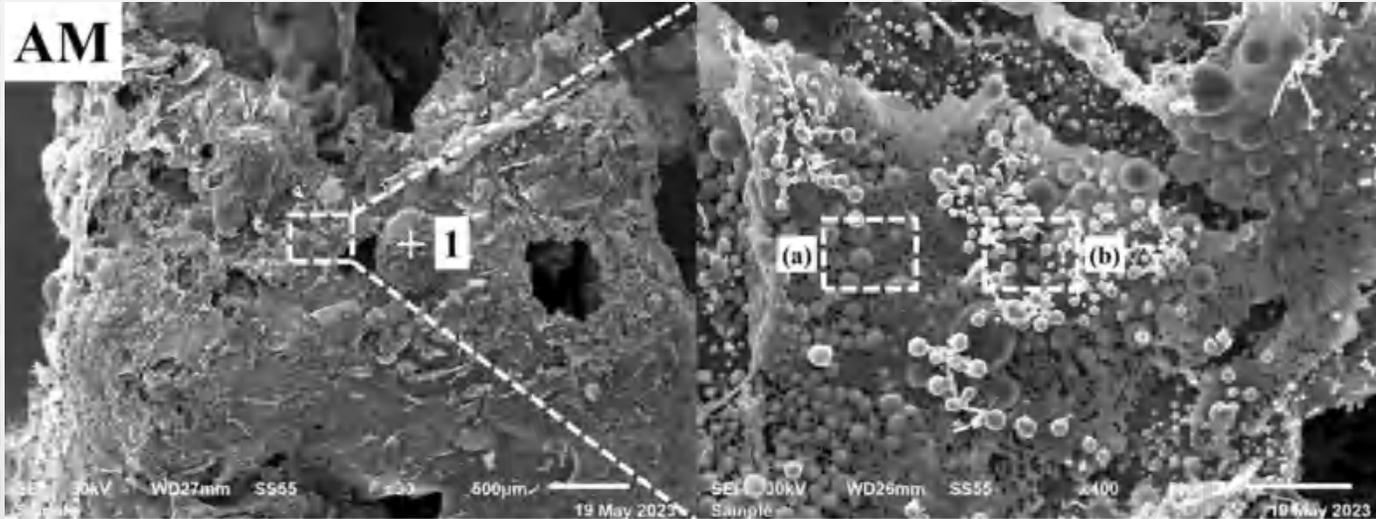
Filter A did not significantly reduce the T.O content, which was possibly related to its fracture and subsequent fall into the molten steel.



# Results and discussion



AM



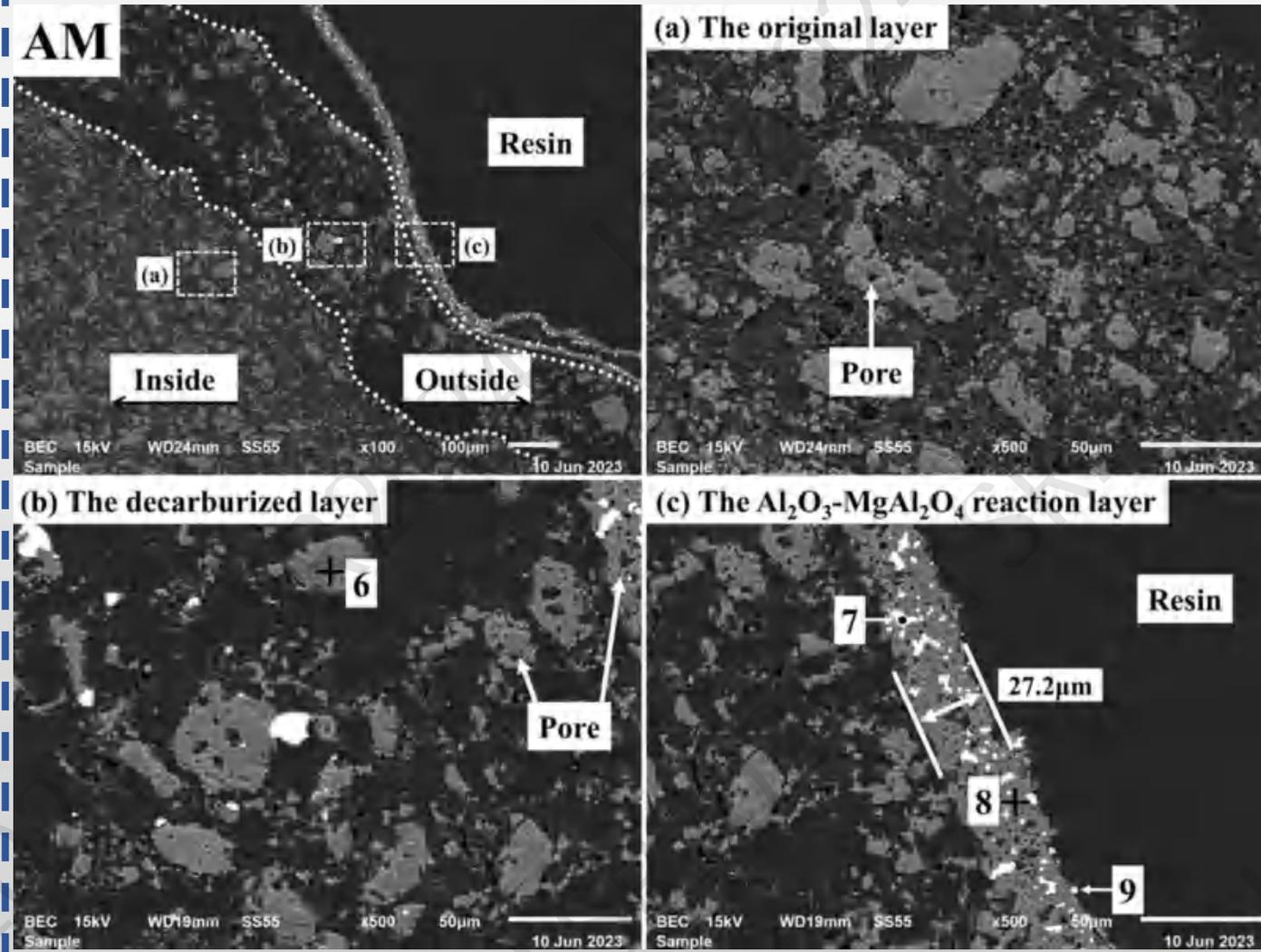
## EDS results (At%)

	O	Mg	Al	Si	Cr	Fe	Possible phases (1600°C)
1	29.81	12.73	55.85			1.61	Corundum, Spinel
2	34.39	10.78	53.76			1.07	Corundum, Spinel
3	18.82	8.83	32.26	13.80	0.98	25.31	Olivine, Corundum, Composite Spinel
4	43.04	9.28	41.74	5.94			Corundum, Spinel, Olivine
5	4.65	7.45	17.79	17.89	1.89	50.33	Fe-Si-Al Alloy, Composite Spinel

$\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  reaction layer was observed on the surface of the filter, exhibiting a significant distribution of micropores across its irregular surface. It is observed that  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  and  $\text{Al}_2\text{O}_3\text{-MgO-SiO}_2$  inclusions in the molten steel were adsorbed on the surface of the porous  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  layer.



# Results and discussion



## EDS results (At%)

	O	Mg	Al	Si	Fe	Possible phases (1600°C)
6	50.22	9.74	40.04			Corundum, Spinel
7			4.60		95.40	Fe-Al Alloy
8	42.59	18.68	38.15		0.58	Composite Spinel
9			2.55	18.92	78.53	Fe-Si-Al Alloy

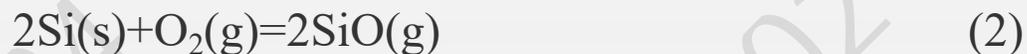
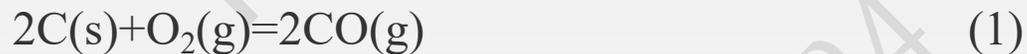
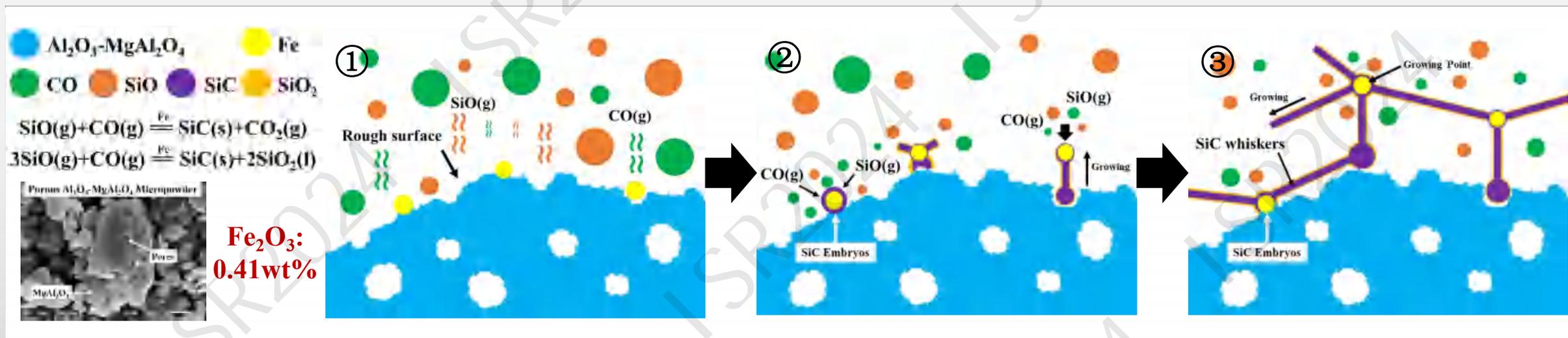
- (a) The original layer, located far from the molten steel, retained its original morphology;
- (b) The decarburized layer, located near the molten steel, **experienced intense carbothermal reduction reactions**, resulting in a loose structure;
- (c) The  $Al_2O_3$ - $MgAl_2O_4$  reaction layer, **in direct contact with the molten steel**, had a uniform thickness of approximately **27.2  $\mu$ m**.



# Results and discussion

## Discussion

Firstly, porous  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  micropowder facilitated the formation of  $\beta\text{-SiC}$  whiskers within the filter.



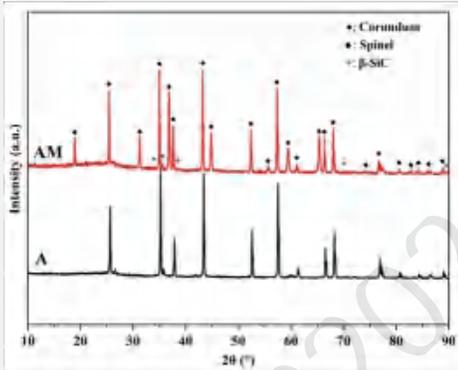
The  $\beta\text{-SiC}$  whiskers branched out and epitaxially grew, originating from **the Fe catalyst droplets, forming a uniformly distributed  $\beta\text{-SiC}$  whisker network.**



# Results and discussion

## Discussion

Secondly, the  $\beta$ -SiC and  $MgAl_2O_4$  phase significantly enhanced the thermal shock resistance of the filter.



Porous  $Al_2O_3$ - $MgAl_2O_4$  micropowder  
(90.01 wt%  $Al_2O_3$ , 8.60 wt%  $MgO$ )

Relative contents of phase (XRD results)

Filter	Corundum	Spinel	$\beta$ -SiC
AM	58 wt%	33 wt%	9 wt%
A	97 wt%	-	3 wt%

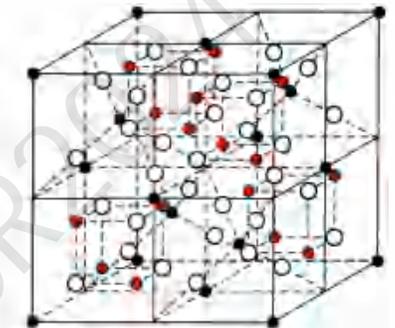
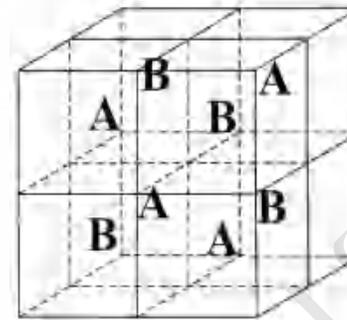
Spinel:

$MgAl_2O_4$

● : Mg

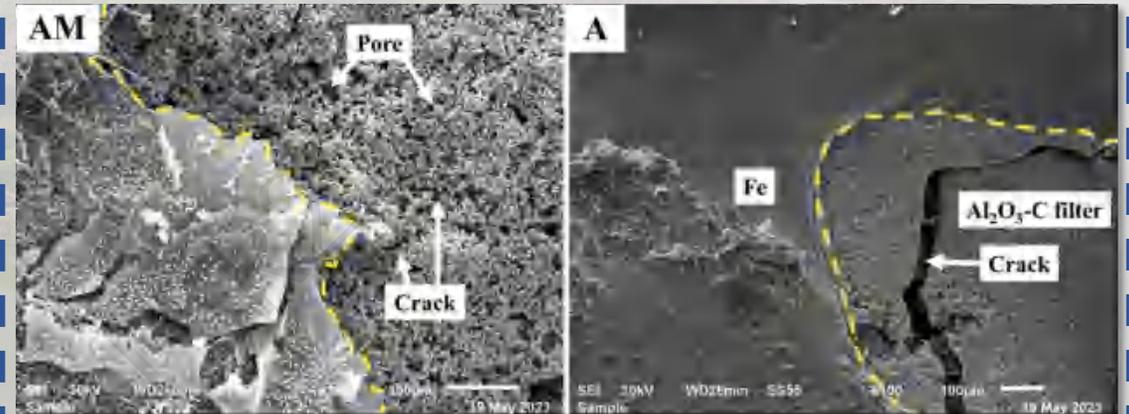
● : Al

○ : O



Filter AM exhibited a retention rate of 91.11% in its cold compressive strength after three thermal shock tests.

The  $\beta$ -SiC whiskers interlocking porous  $Al_2O_3$ - $MgAl_2O_4$  and carbon particles in the filter AM prevented the crack propagation during the immersion test, so that no large cracks occurred in the filter AM.





# Results and discussion

## Discussion

Thirdly, the combination of the **microporous structure and  $MgAl_2O_4$  phase** synergistically improved the **purification efficiency** of the filter for molten steel.

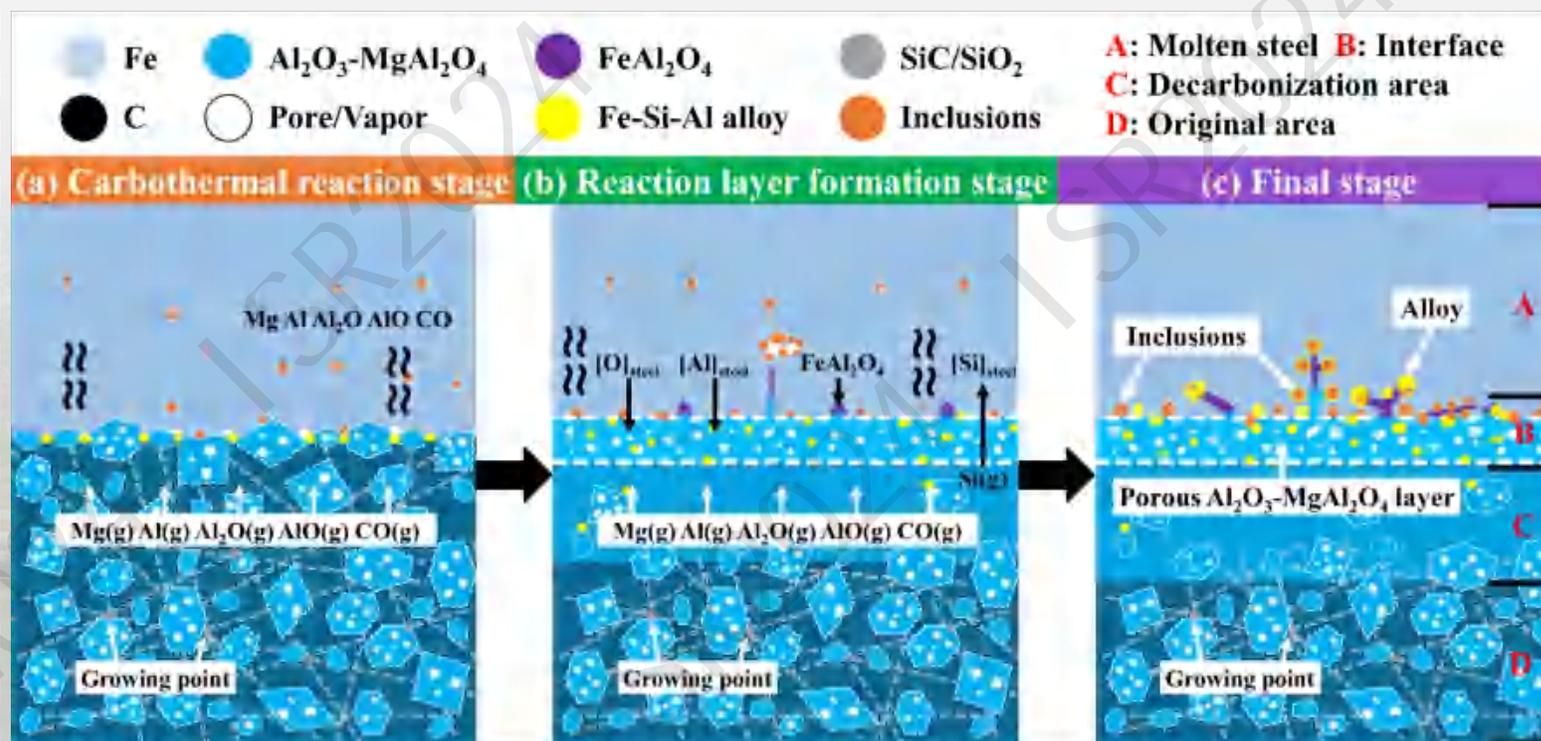
High purity argon

$O_2$  is less than 1ppm

$$P(O_2) \leq 1 \cdot 10^{-1} \text{ Pa}$$

Based on the experimental conditions, the process of interfacial reactions between filter AM and the molten steel will be discussed from **the carbothermal reaction stage, the reaction layer formation stage and the final stage.**

### Schematic illustration of immersion test using filter AM





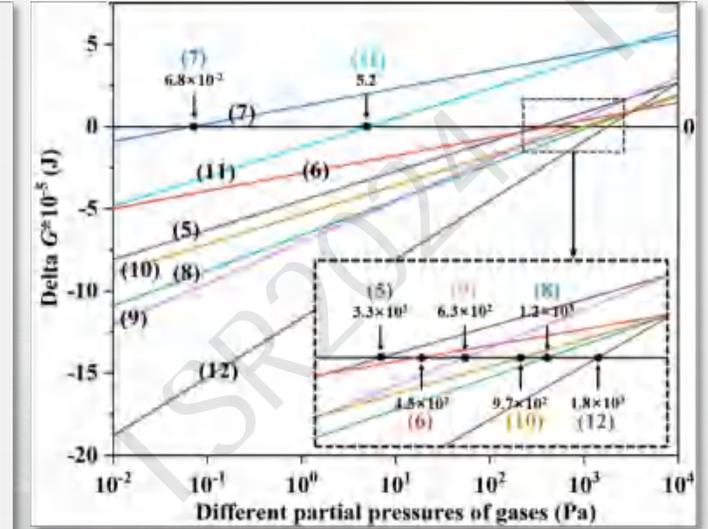
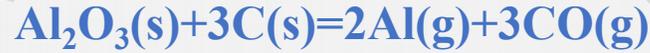
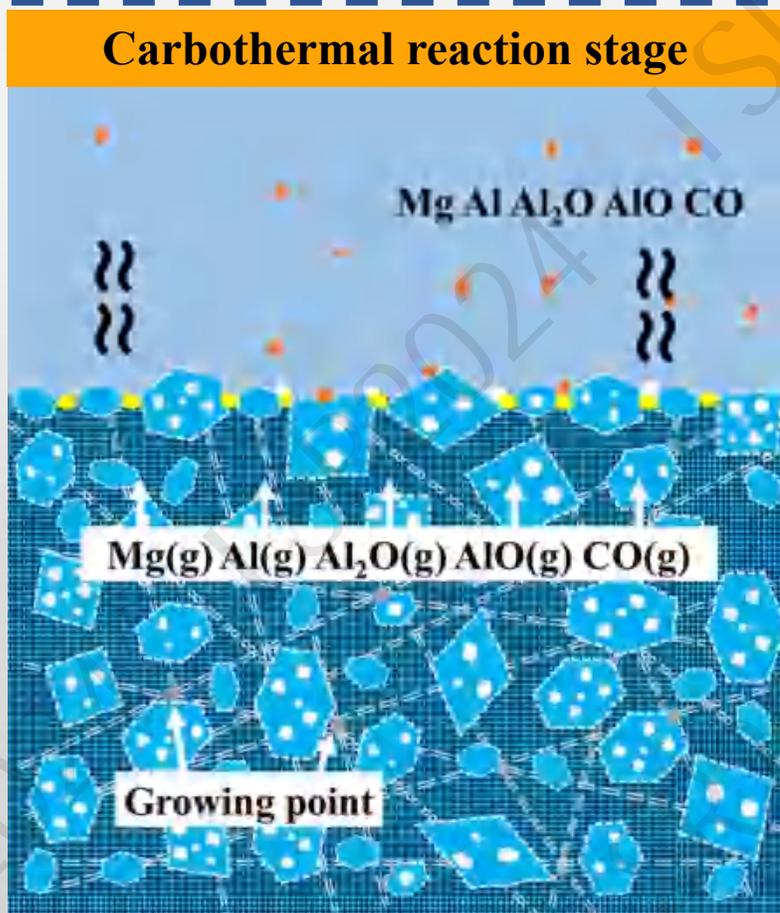
# Results and discussion



## Discussion

### (a) Carbothermal reaction stage

#### Carbothermal reaction stage



Al (g):  $6.3 \times 10^2$  Pa

$\text{Al}_2\text{O}$  (g):  $9.7 \times 10^2$  Pa

AlO (g): 5.2 Pa

Mg (g):  $1.8 \times 10^3$  Pa

CO (g):  $1.8 \times 10^3$  Pa

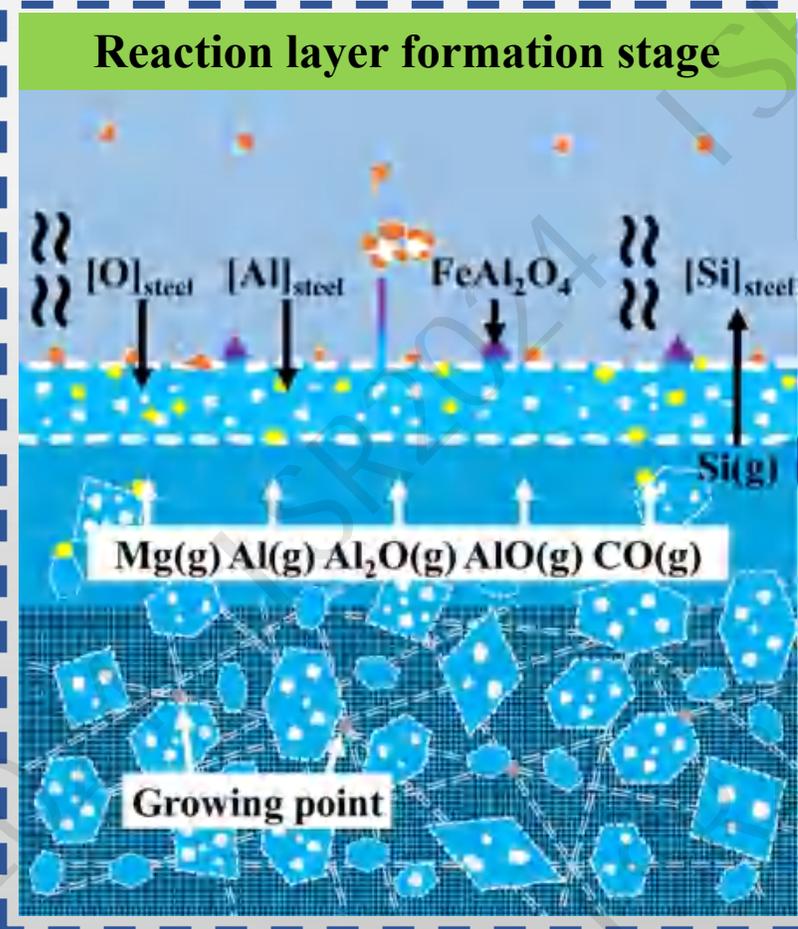


# Results and discussion

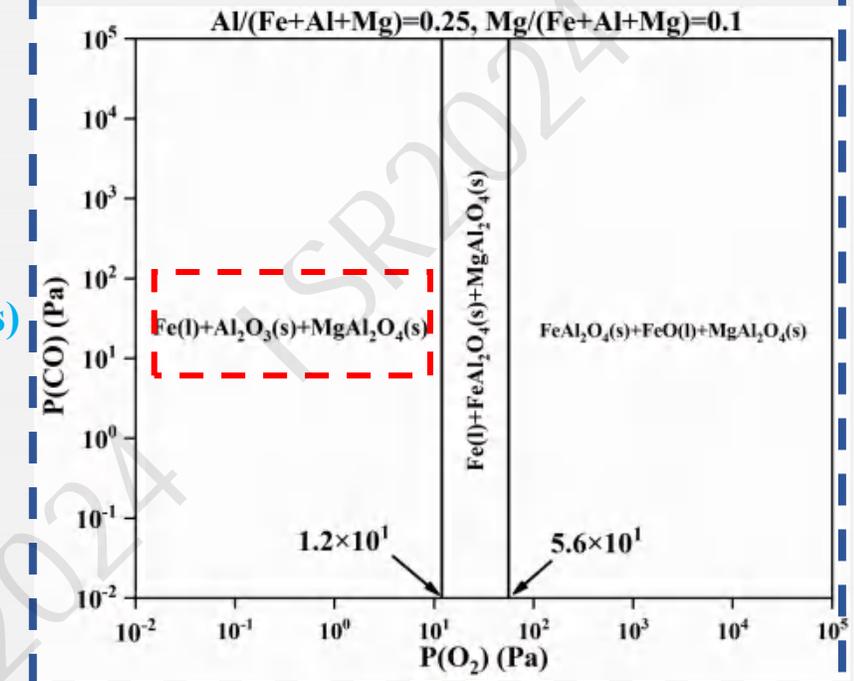


## Discussion

### (b) Al<sub>2</sub>O<sub>3</sub>-MgAl<sub>2</sub>O<sub>4</sub> reaction layer formation stage



Predominance area diagram of the Al-Mg-Fe-O-C system at 1600°C



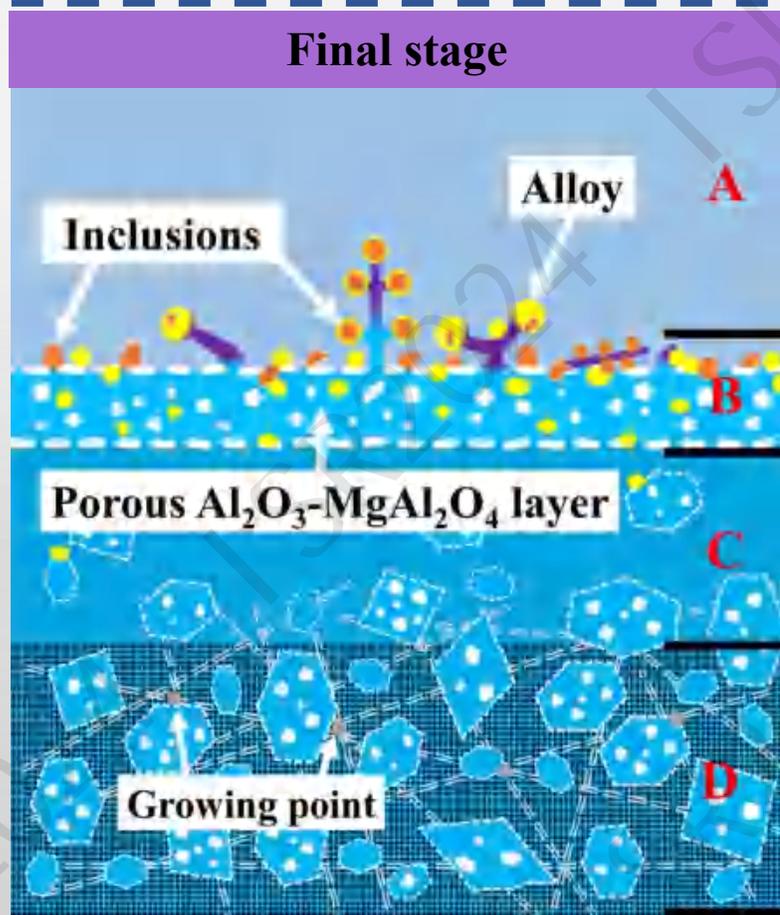
**FeAl<sub>2</sub>O<sub>4</sub> is the stable phase** when the partial pressure of O<sub>2</sub> is **higher than 1.2 × 10<sup>1</sup> Pa** and **Al<sub>2</sub>O<sub>3</sub> is the stable phase** when the partial pressure of O<sub>2</sub> is **lower than 1.2 × 10<sup>1</sup> Pa**.



# Results and discussion

## Discussion

### (c) Final stage



A:  
Molten steel

B:  
Interface

C:  
Decarbonization area

D:  
Original area

The filter AM had a rougher surface structure, where the aforementioned reaction process led to **the formation of the  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  layer and  $\text{FeAl}_2\text{O}_4$  nuclei**. As the inclusions were continuously and gradually adsorbed, **the  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  layer progressively thickened, accompanied by the formation of acicular composite spinel on the surface**. The Fe on the surface dissolved the Al and Si vapors to form Fe-Si-Al alloy spheres, **which encapsulated the inclusions on the filter surface and linked with the porous layer through acicular composite spinel**.

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— PART FOUR —



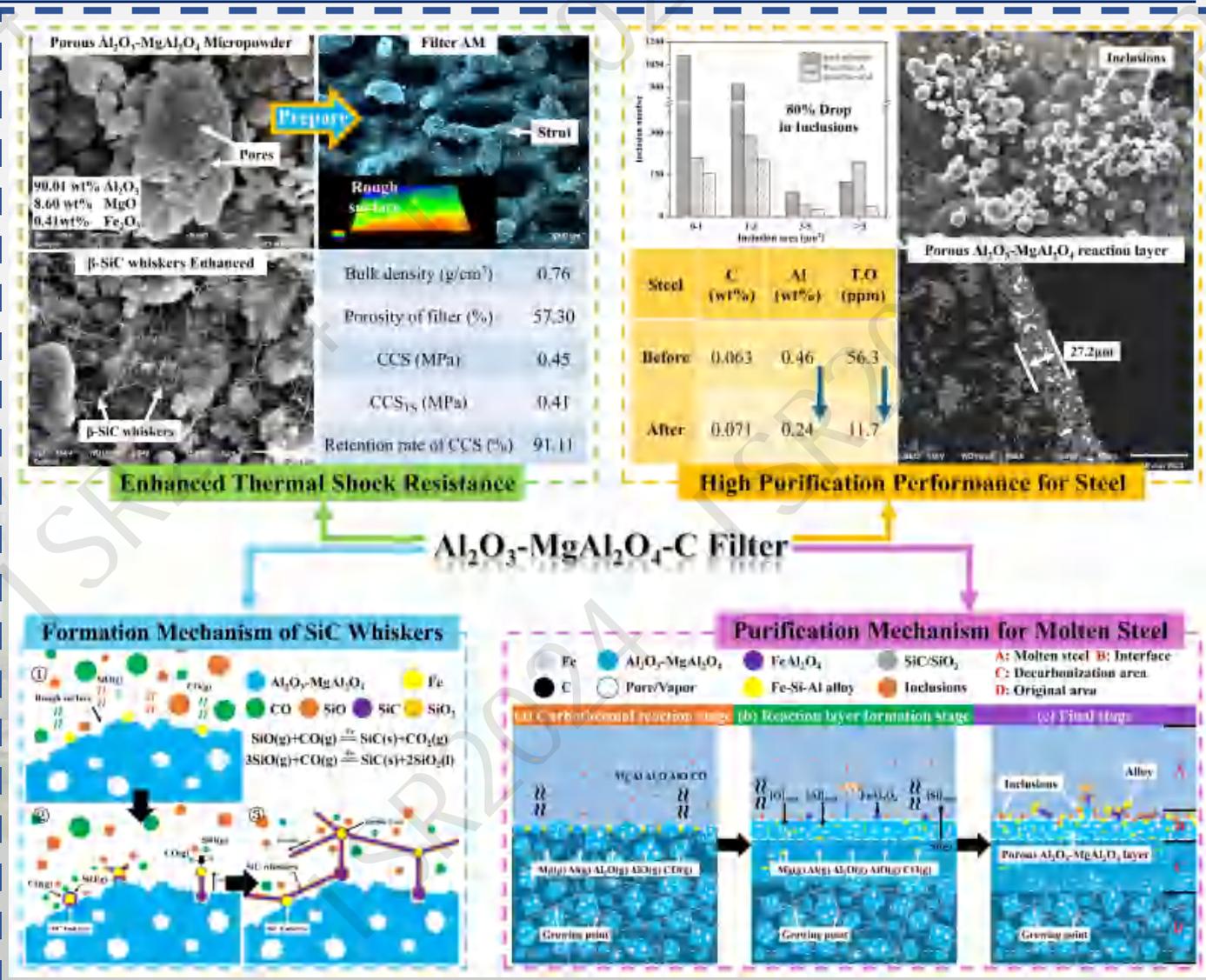
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# Conclusions



# Conclusions

- ◆ The porous  $\text{Al}_2\text{O}_3\text{-MgAl}_2\text{O}_4$  micropowder contained more  $\text{Fe}_2\text{O}_3$ . With the progress of VLS mechanism,  $\beta\text{-SiC}$  embryos formed on the rough grains surface under the catalysis of Fe, **leading to the development of a network structure of  $\beta\text{-SiC}$  whiskers.**
- ◆ The  $\beta\text{-SiC}$  and  $\text{MgAl}_2\text{O}_4$  phase significantly enhanced the **thermal shock resistance** of the filter. After three thermal shock tests, the cold compression strength of filter AM (0.41 MPa) was higher than that of filter A (0.36 MPa), with a strength retention rate as high as 91.11%.
- ◆ The combination of the **microporous structure and  $\text{MgAl}_2\text{O}_4$**  phase synergistically improved the **purification efficiency** of the filter for molten steel.



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**Thank you for your listening!**

PLEASE GIVE ME AS MUCH CRITICISM AS POSSIBLE

**Jinwen Song | Supervisor: Prof. Wen YAN**